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## Phase transitions in bismuth under rapid compression\*

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The structural phase transitions of bismuth under rapid compression has been investigated in a dynamic diamond anvil cell using time-resolved synchrotron x-ray diffraction. As the pressure increases, the transformations from phase I, to phase II, to phase III, and then to phase V have been observed under different compression rates at 300 K. Compared with static compression results, no new phase transition sequence appears under rapid compression at compression rate from 0.20 GPa/s to 183.8 GPa/s. However, during the process across the transition from phase III to phase V, the volume fraction of product phase as a function of pressure can be well fitted by a compression-rate-dependent sigmoidal curve. The resulting parameters indicate that the activation energy related to this phase transition, as well as the onset transition pressure, shows a compression-rate-dependent performance. A strong dependence of over-pressurization on compression rate occurs under rapid compression. A formula for over-pressure has been proposed, which can be used to quantify the over-pressurization.

**Keywords:** bismuth, high pressure, rapid compression, over-pressurization**PACS:** 62.50.-p, 78.47.D-, 61.05.cp, 07.35.+k**DOI:** 10.1088/1674-1056/28/3/036201

## 1. Introduction

Bismuth (Bi) has been widely studied experimentally and theoretically under high pressure due to its complex phase diagram and its abundant pressure-induced polymorphic phases.<sup>[1–20]</sup> Previous studies have indicated that bismuth undergoes three pressure-induced structural phase transitions in the pressure range below 10 GPa at room temperature. Under ambient conditions, bismuth is stable in phase I (Bi-I) having a rhombohedral A7-type crystal structure with space group  $R\bar{3}m$ .<sup>[1]</sup> The first phase transition occurs at around 2.5 GPa to phase II (Bi-II),<sup>[2]</sup> which has a monoclinic structure with space group  $C12/m1$ .<sup>[3]</sup> Bi-II has a very narrow stability pressure range, and it transforms to the complex phase III (Bi-III) as the pressure increases to 2.7 GPa.<sup>[4]</sup> Bi-III has been reported as a composite tetragonal host-guest structure.<sup>[5]</sup> Bi-III transforms to phase V (Bi-V) at 7.7 GPa,<sup>[6]</sup> which is a body-centered cubic structure shown to be stable up to 222 GPa.<sup>[7]</sup> In addition to the static high pressure investigations, shock compression studies of bismuth have also been reported on shock-induced transition and melting.<sup>[8–18]</sup> The three structural transitions have been reported in shock compression with the phase transition pressures at 2.7 GPa,<sup>[8]</sup> 3.2 GPa,<sup>[15]</sup> and 7.0 GPa,<sup>[10]</sup> which are different from the results in static compressions. Furthermore, a high temperature and high pressure

phase of bismuth (Bi-IV) with an orthorhombic structure has been identified at  $\sim 4$  GPa and  $\sim 500$  K.<sup>[19]</sup>

Up to now, the pressure-induced structure transformation in bismuth has mainly been studied using static compression in diamond anvil cell (DAC) or shock wave loading. However, the kinetics of a pressure-induced phase transition, e.g., transition pressure and characteristic time, should be related to the pressure loading rate. Recently, the development of dynamic diamond anvil cells (dDAC) with time-resolved synchrotron x-ray diffraction make structural measurement possible under rapid compression with timescales in the order of a millisecond.<sup>[22]</sup>

In this paper, we present a study of structural phase transitions in bismuth under rapid compression at room temperature using the fast loading DAC technique with time-resolved synchrotron x-ray diffraction, which has been recently developed at the Beijing Synchrotron Radiation Facility (BSRF). The experimental results show that Bi-I, Bi-II, Bi-III, and Bi-V appear one by one as the pressure increases from ambient pressure to  $\sim 12$  GPa. There is no change in the phase transition path of bismuth under rapid compression at different compression rates (0.20–183.8 GPa/s) compared with the static high pressure experiments. In the phase transition process from Bi-III to Bi-V, the increasing volume fraction of Bi-V versus pres-

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sure generally gives a sigmoidal curve which can be well fitted by a compression-rate-dependent function. The fitting results show that, as the compression rate increases, the activation energy associated with this transformation decreases while the onset transition pressure increases. Moreover, a strong compression rate dependence of over-pressurization occurs under rapid compression, and a formula of over-pressure is defined to quantify over-pressurization.

## 2. Experimental details and theoretical methods

A symmetric diamond anvil cell equipped with a pair of diamond anvils with a culet diameter of 250  $\mu\text{m}$  was used for the rapid compression experiments. A rhenium (Re) gasket was used with the sample chamber diameter of  $\sim 100 \mu\text{m}$  and pre-indented thickness of  $\sim 40 \mu\text{m}$ . Bismuth sample powder with 99.5% purity (Alfa Aesar) was pressed into a thin slice, and then loaded into the sample chamber. Sodium chloride (NaCl, from Alfa Aesar) was added as the pressure medium to obtain quasi-hydrostatic pressure conditions, as well as the internal pressure marker. After sample loading, the DAC was inserted into a dDAC which was driven by a piezoelectric actuator.

Time-resolved synchrotron x-ray diffraction experiments were carried out using the angle-dispersive diffraction method with a monochromatic x-ray beam wavelength of 0.6199  $\text{\AA}$  at the 4W2 high pressure station at BSRF. The incident x-ray beam was focused in the horizontal and vertical directions to a  $25 \times 10 \mu\text{m}^2$  (full width at half maximum (FWHM)) spot on the sample using Kirkpatrick–Baez mirrors.<sup>[21]</sup> The two-dimensional diffraction pattern was collected using a PILATUS2 2M area detector with an exposure period of 100 ms and an exposure time of 90 ms for each image.

A function generator and a digital oscilloscope were used for synchronization, triggering, and monitoring the timing between the x-ray detector and the dDAC rapid compression device. We used trapezoid waveforms to trigger the piezoelectric actuator to modulate the rapid compressions in the dDAC, and used pulse waveform to trigger the detector to collect the diffraction patterns. By adjusting the timing of the trigger waveforms of the dDAC and detector, the pressure compression and the x-ray diffraction image collection proceeded simultaneously. The Fit2D program<sup>[23]</sup> was used to integrate the diffraction images to obtain intensity versus 2-theta plots. Then the diffraction data were analyzed to index the diffraction peaks by using the Jade (MDI Jade 6) program. The sample pressure was determined from the NaCl (200) diffraction peak in each diffraction image by using the equation of state (EOS) previously reported.<sup>[24]</sup> The compression rate of the experiment was estimated by the rising slope of the determined pressure versus time.

In the case of rapid compression, the volume fraction of product phase during the phase transformation process as a

function of pressure can be described by Eq. (1)<sup>[25]</sup> which is based on the Johnson–Mehl–Avrami–Kolmogorov (JMAK) equation

$$f(P) = 1 - \exp[-x_{\text{ex}}(P)], \quad (1)$$

$$x_{\text{ex}}(P) = k^{-1-n} \int_{P_0}^P I(P) \left[ \int_{P'}^P v(P'') dP'' \right]^n dP', \quad (2)$$

$$I(P) = I_0 \exp(-Q_N/k_B T), \quad (3)$$

$$v(P) = v_0 \exp(-Q_G/k_B T) [1 - \exp(\Delta G_V/k_B T)], \quad (4)$$

where  $x_{\text{ex}}(P)$  is the extended volume, nucleation rate  $I(P)$  and growth rate  $v(P)$  are pressure-dependent at a given compression rate  $k$ ,  $Q_N$  and  $Q_G$  are the activation energies in the nucleation and growth processes, respectively,  $\Delta G_V$  is the difference of the Gibbs free energy induced by the volume change associated with the transformation,  $P_0$  is the onset phase transition pressure,  $n$  is the growth exponent,  $k_B$  is the Boltzmann constant, and  $T$  is the room temperature. To the zero-order approximation, we assume that the nucleation rate  $I(P)$  and growth rate  $v(P)$  change very little with pressure. The equations (1)–(4) are reduced to

$$f(P) = 1 - \exp \left[ -A \left( \frac{P - P_0}{k} \right)^{n+1} \right], \quad (5)$$

where

$$A = \frac{I_0 v_0^n}{n+1} \times \exp \left( -\frac{Q_{N0} + nQ_{G0}}{k_B T} \right). \quad (6)$$

Here,  $A$  is related to the activation energy at a given compression rate. The dimensionality of the growth  $n$ , the so-called Avrami exponent, should be equal to 3 in this study because nuclei form in the grain interior and grow in three dimensions.<sup>[26]</sup> The compression rate  $k$  can be obtained by fitting the pressure–time data. There are only two parameters  $A$  and  $P_0$  in fitting the volume fraction data according to Eq. (5).

## 3. Results and discussion

Figure 1 shows the pressure dependence of the interplanar distances of Bi under different compression rates. Bi-I is the initial phase before each rapid compression. As the pressure increases, the decreases and transformations of the interplanar distances have been clearly shown in Fig. 1, indicating the structural evolutions of bismuth from phase I, to phase II, to phase III, and then to phase V induced by rapid compressions. In the cases of the compression rates of 92.0 GPa/s and 183.3 GPa/s, Bi-II has not been observed due to the very narrow stability pressure range of phase II and an insufficient time resolution in the experiments. Obviously, no new phase transition sequence occurs during rapid compressions with the compression rate below 183.8 GPa/s.



cesses of phase transitions induced by temperature under heating at given heating rates.<sup>[27-29]</sup> Similarly, the differentiation of Eq. (5) results in a transformation rate equation under rapid compression condition

$$\frac{df}{dP} = (n+1) \frac{A^{1/(n+1)}}{k} (1-f)(-\ln(1-f))^{n/(n+1)}, \quad (7)$$

where parameter  $A$  dependent on  $k$  is related to the activation energy associated with this transition, as mentioned above. From Eq. (7), it can be seen that the transformation rate only depends on the pressure (through  $k$ ) and the transformed fraction. At a given pressure  $P$ , the change of transformed fraction in the small range of  $\Delta P$  can be expressed as

$$\Delta f = \frac{df}{dP} \times \Delta P. \quad (8)$$

Actually,  $\Delta f$  can be treated as a weighting factor for characterizing the contribution of the current pressure to the phase transition. Here, we define the over-pressure  $P_{ov}$  as a weighted average pressure which can be used to quantify over-pressurization. It may be described by

$$P_{ov} = \int P \times \frac{df}{dP} dP, \quad (9)$$

where the integration is over the pressure range of the phase transition. As shown in Fig. 4, as the compression rate increases, the over-pressure increases as well as the onset transition pressure.

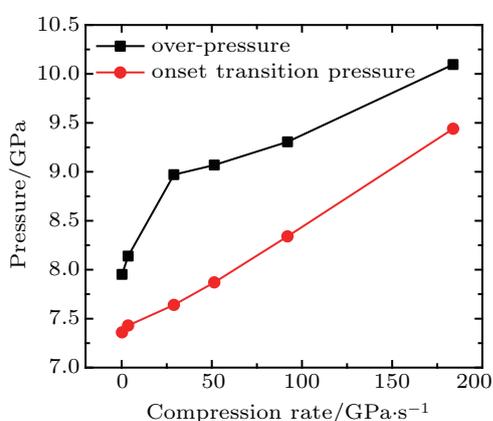


Fig. 4. The influence of compression rate on the over-pressure and onset transition pressure.

#### 4. Conclusions

In summary, using a dynamic-DAC and time-resolved synchrotron x-ray diffraction, the structural evolution of bismuth induced by rapid compression has been probed with

100 ms time resolution at room temperature. The structural evolutions of bismuth induced by rapid compressions in the rate range from 0.2 GPa/s to 183.3 GPa/s have been observed. The phase transition sequence is the same as that in static high pressure experiments. In the process from Bi-III to Bi-V, the volume fraction of Bi-V versus pressure can be well fitted by a sigmoidal curve. The fitting results show that the activation energy associated with this transformation depends on the compression rate. A strong dependence of over-pressurization on compression rate has been found under rapid compression. In order to quantify the over-pressurization, a formula for over-pressure has been proposed.

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